

MODELING POWDER X-RAY DIFFRACTION PATTERNS OF THE CLAY MINERALS SOCIETY KAOLINITE STANDARDS: KGa-1, KGa-1b, AND KGa-2

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Abstract—Three kaolinite reference samples identified as KGa-1, KGa-1b, and KGa-2 from the Source Clays Repository of The Clay Mineral Society (CMS) are used widely in diverse fields, but the defect structures have still not been determined with certainty. To solve this problem, powder diffraction patterns of the KGa-1, KGa-1b, and KGa-2 samples were modeled. In a kaolinite layer among three symmetrically independent octahedral sites named as A, B, and C and separated from each other by $b/3$ along the b parameter, the A and B sites are occupied by Al cations, whereas, the C sites located along the long diagonal of the oblique kaolinite unit cell are vacant. The layer displacement vectors \mathbf{t}_1 and \mathbf{t}_2 are related by a pseudo-mirror plane from defect-free 1Tc kaolinite enantiomorphs, whereas, the random interstratification within individual kaolinite crystallites creates right-hand and left-hand layer sub-sequences producing structural disorder. A third layer displacement vector, \mathbf{t}_0 , located along the long diagonal of the oblique layer unit cell that contains the vacant octahedral site and coincides with the layer pseudo-mirror plane may exist. Thus, a structural model should be defined by the probability of \mathbf{t}_1 , \mathbf{t}_2 , and \mathbf{t}_0 layer displacement translations $W\mathbf{t}_1$, $W\mathbf{t}_2$, and $W\mathbf{t}_0$, respectively, determined by simulated experimental X-ray diffraction (XRD) patterns. X-ray diffraction patterns were calculated for structures with a given content of randomly interstratified displacement vectors, and other XRD patterns were calculated for a physical mixture of crystallites having contrasting structural order with only C-vacant layers. The samples differ from each other by the content of high- and low-ordered phases referred to as HOK and LOK. The HOK phase has an almost defect-free structure in which 97% of the layer pairs are related by just the layer displacement vector \mathbf{t}_1 and only 3% of the layer pairs form the enantiomorphic fragments. In contrast, the LOK phases in the KGa-1, KGa-1b, and KGa-2 samples differ from HOK phases by the occurrence probabilities for the \mathbf{t}_1 , \mathbf{t}_2 , and \mathbf{t}_0 layer displacements. In addition, the LOK phases contain stacking faults that displace adjacent layers in arbitrary lengths and directions. Low XRD profile factors ($R_p = 8\text{--}11\%$) support the defect structure models. Additional structural defects and previously published models are discussed.

Key Words—Computer simulation, Defect structure, Kaolinite, X-ray diffraction.

INTRODUCTION

Three kaolinite reference samples identified as KGa-1, KGa-1b, and KGa-2 from the Source Clays Repository of The Clay Mineral Society are widely used by clay researchers. These kaolinite samples are representative of commonly occurring, natural kaolinite materials and are distinguished by phase purity and availability. The KGa-1 and KGa-1b samples, often referred to as well ordered or well crystallized kaolinite varieties, were collected from close geographical locations and stratigraphic positions in Georgia (Van Olphen and Fripiat, 1979; Pruett and Webb, 1993; Moll, 2001). The X-ray diffraction (XRD) patterns of KGa-1 and KGa-1b are almost identical and both samples have a similar chemical composition (Pruett and Webb, 1993; Chipera and Bish, 2001; Mermut and Cano, 2001). The KGa-2 standard is also from Georgia (Van Olphen and Fripiat, 1979; Pruett and Webb, 1993; Mermut and Cano, 2001) and has a disordered structure. These kaolinite standards

are found in relatively pure deposits, but contain trace quantities of anatase, Fe-oxides, quartz, crandallite, micas, and smectites.

The pure phase composition of KGa-1, KGa-1b, and KGa-2 provides an opportunity for an unambiguous interpretation of experimental data obtained by different methods to investigate the behavior of samples subjected to different treatments and reactions, and to reveal the sensitivity of ordered and disordered kaolinite samples to the same treatments and reactions. For example, these kaolinite standards were used to assign vibration bands in the middle- and near-infrared regions of the infrared (IR) spectra (Madejova and Komadel, 2001; Johnston *et al.*, 1990, 2008; Balan *et al.*, 2001, 2010). Paris (2014) recorded for the first time the ^{27}Al nuclear magnetic resonance (NMR) spectrum of the KGa-1b kaolinite sample at low magnetic field (7T). The analysis of the spectrum allowed determination of the isotropic chemical shift and the quadrupolar interaction parameters. The combination of these data with first principle calculations of the electric field gradient shows that the two adjacent octahedral aluminum sites have different geometry. The KGa-1, KGa-1b, and KGa-2 samples were used to determine kaolinite surface charge proper-

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ties (Schroth and Sposito, 1997; Wu, 2001). Investigation of dissolution rates of the KGa-1b and KGa-2 samples in oxalic acid and inorganic acids shows that the fundamental structure of the kaolinites, rather than specific surface details, exerts the greater influence on dissolution kinetics (Sutheimer *et al.*, 1999). The KGa-1b and KGa-2 kaolinites were used to study the sorption of uranium (U) to kaolinite from water and more U(VI) was sorbed by KGa-2 than by KGa-1b (Bachmaf and Merkel, 2011). Investigating effects of ultrasound on KGa-1 kaolinite, Franco *et al.* (2004) found that this treatment did not cause amorphization, but introduced stacking faults and formed kaolinite fragments that were enantiomorphic to the main kaolinite matrix.

Different methods and approaches were used to study the transformation of the KGa-1 and KGa-2 kaolinite samples into metakaolinite. Based on the evolution of time-resolved, energy-dispersive powder diffraction patterns of KGa-1 and KGa-2 samples, Bellotto *et al.* (1995) investigated the influence of structural defects on the dehydroxylation reaction. Drits and Derkowski (2015), using the multi-cycle heating and cooling thermogravimetric method, discovered the kinetic behavior of defect-free Keokuk, KGa-1, and KGa-2 samples during partial dehydroxylation. Frost and Vassallo (1996) compared the dehydroxylation reactions of kaolinite KGa-1, dickite, and halloysite using infrared emission spectroscopy and showed that the kaolinite sample lost the inner- and inner-surface hydroxyls simultaneously, whereas, dickite and halloysite lost the inner-surface hydroxyls before inner-hydroxyls.

Recently, the structure of kaolinite has been studied using modern *ab initio* computation methods. White *et al.* (2010), using the density functional theory (DFT) and a step-wise dehydroxylation approach, simulated the local structure of KGa-1b at different stages of partial dehydroxylation. Analysis of the experimental inelastic neutron scattering (INS) spectrum obtained for KGa-1b in combination with DFT analysis were used to determine hydrogen coordinate positions in the kaolinite structure (White *et al.*, 2009, 2013a, 2013b; Sperinck *et al.*, 2011). To reconcile the experimental INS data with the DFT analysis, White *et al.* (2013a) concluded that the specific layer displacement vector $\mathbf{t}_0 = -0.3151a - 0.3151b$ is the dominant stacking fault type among other stacking faults present in the KGa-1b structure.

Application of inelastic neutron scattering to analyze the location and dynamics of hydrogen in partially or completely dehydroxylated KGa-1b kaolinite indicated a preferential loss of inner surface H-atoms during partial dehydroxylation, and the residual hydrogen atoms in metakaolinite were located in diverse bonding environments. As with natural KGa-1b kaolinite, the INS spectrum of the partially dehydroxylated sample correlated with the INS spectrum calculated for the defect model containing the specific layer displacement, \mathbf{t}_0 , and

consisting of the contributions from inner- and inner-surface H-atoms (White *et al.*, 2013b).

These mentioned studies are only part of the many publications in which KGa-1, KGa-1b, and KGa-2 were used in diverse scientific, industrial, and other applied fields. A common feature of these and other numerous publications in which these kaolinite samples were used, however, is that the authors do not know the nature of the actual defect structure of the studied kaolinite varieties. For example, the KGa-2 sample is assumed to have structural disorder that is related only to a high proportion of stacking faults (Bookin *et al.*, 1989; Plançon *et al.*, 1989). In contrast, in KGa-1 and KGa-1b, the nature of the structural defects is ambiguous. Stacking faults in the KGa-1 and KGa-1b structure may be distributed either in a mono-phase (Artioli *et al.*, 1995) or in two discrete phases containing contrasting amounts of stacking faults (Plançon *et al.*, 1989).

Because of uncertainties related to the crystal structure of KGa-1b and KGa-2 kaolinite samples, various authors were free to apply models that helped them to interpret their experimental results. Because kaolinite occurs as fine particles, X-ray diffraction is a most convenient and useful technique for crystallographic characterization including stacking disorder. The actual defect structure of kaolinite, however, has apparently not been studied by this method during the last 20 years (Plançon and Tchoubar, 1977; Bookin *et al.*, 1989; Plançon *et al.*, 1989, 1990; Artioli *et al.*, 1995). During this period, significant progress was achieved in modeling the experimental powder XRD patterns of various highly dispersed and disordered structures of layered minerals (Drits *et al.*, 1997a, 2002a,b, 2004, 2011a,b; Ferrage *et al.*, 2007; Lanson *et al.*, 2009; Lindgreen *et al.*, 2002; Plançon *et al.*, 1989; Sakharov *et al.*, 1999, 2004; McCarty *et al.*, 2009). The main task of this paper was to determine the actual defect structure of the CMS kaolinite standards KGa-1, KGa-1b, and KGa-2 by modeling powder XRD patterns.

KAOLINITE DEFECT STRUCTURE (REVIEW)

Kaolinite, $\text{Al}_2\text{Si}_2\text{O}_5(\text{OH})_4$, is a dioctahedral 1:1 layer mineral and the structure consists of one octahedral sheet and one tetrahedral sheet that are bound to each other *via* tetrahedral sheet apical oxygen atoms. Strong cohesion of the adjacent layers is formed by hydrogen bonding from OH groups on the basal surface of one layer to oxygen atoms forming a basal surface of an adjacent layer. The octahedral sheet of the kaolinite layer contains three possible octahedral cation sites differing in the arrangement of OH groups and oxygen atoms, which are labeled A, B, or C (Figure 1) and the 1:1 layers will hereafter be identified and designated by the letter corresponding to the type of vacant site (Bailey, 1963). In the conventional right-hand kaolinite unit cell, the tetrahedral sheet is at the base of each 1:1

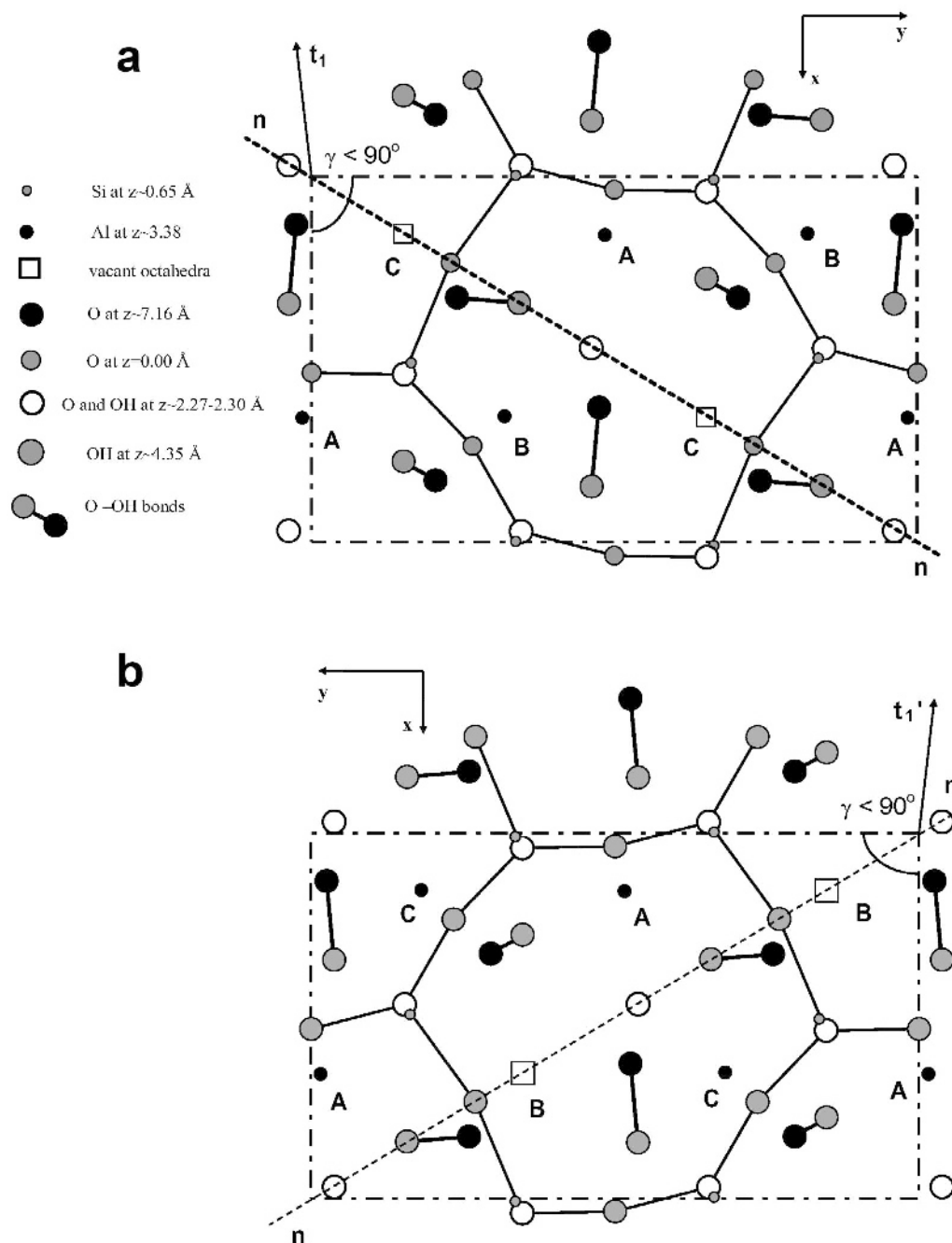


Figure 1. Mutual arrangement of a kaolinite layer; atoms projected on the (001) plane based on the Cartesian atomic coordinates refined by Bish and von Dreele (1989). Possible octahedral sites are labeled A, B, and C. Small black circles indicate octahedral sites occupied by Al, whereas, light squares indicate vacant C sites (a) and vacant B sites (b). Basal oxygen atoms (large black circles) of the upper kaolinite layer paired with basal surface OH groups (small gray circles) of the lower kaolinite layer. The second layer is shifted by t_1 (a) and t'_1 (b) relative to the lower layer. The symbolic notation of kaolinite layer atoms and z -positions are shown (a).

layer and the vacant C site is located along the long diagonal of the unit cell (Bailey, 1963; 1988). Figure 1a shows a fragment of a C vacant layer built using atomic coordinates refined by Bish and von Dreele (1989). A defect-free kaolinite has a one-layer triclinic structure

(Brindley and Nakahira, 1958; Bish and von Dreele, 1989).

A characteristic feature of most kaolinite varieties is a defect structure (Brindley *et al.*, 1986; Bailey, 1988). The nature of defects in kaolinite structures has been the

subject of intense discussion for a long time. Bookin *et al.* (1989) carried out a comprehensive analysis of possible defects in the kaolinite structure and concluded that the distortions of a real 1:1 layer structure do not allow stacking faults such as $\pm 120^\circ$ mutual layer rotations, $\pm b/3$ layer translations, and the vacancy displacement model suggested in the literature (Murray, 1954; Brindley and Robinson, 1946; Plançon and Tchoubar, 1977). According to Bookin *et al.* (1989), the two-dimensional layer periodicity of defect-free $1T_C$ kaolinite can be described equally well by the orthogonal cell $\{a_0, b_0, \gamma_0\}$ ($\gamma_0 = 90^\circ$) as well as by the two enantiomorphic oblique cells $\{a_1, b_1, \gamma_1\}$ and $\{a_2, b_2, \gamma_2\}$, which are related to each other by a mirror plane passing through the vacant octahedral site and the center of the ditrigonal ring of the tetrahedral sheet in the kaolinite layer and contains the a_0 axis. The numerical values for the oblique cell coincide with those determined for the defect-free kaolinite sample from Keokuk (Suitch and Young, 1987; Bish and von Dreele, 1989), indicating that the obliquity ($\gamma_{1,2} \neq 90^\circ$) of the refined cell of kaolinite is just the choice of the layer unit cell, and the interlayer stacking sequence does not disturb the layer symmetry.

On the other hand, two oblique layer unit cells and the layer displacement vectors \mathbf{t}_1 and \mathbf{t}_2 , which correspond to individual enantiomorphs and are related to each other by the mirror plane (Figure 2), form the same defect-free $1T_C$ kaolinite. A random interstratification of \mathbf{t}_1 (approximately $-a_1/3$) and \mathbf{t}_2 (approximately $a_1/6 - b_1/6$, Figure 2) vectors within individual kaolinite crystallites creates right-hand and left-hand kaolinite structural fragments consisting of the layers with the same vacancy type and, thus, produce most of the structural disorder in kaolinite. Because of the approximately trigonal symmetry of the kaolinite layer, a third layer displacement vector, \mathbf{t}_0 , located along the mirror plane, may also exist in the kaolinite structure. An accurate model of the defect kaolinite structure should, thus, be defined by the probability of \mathbf{t}_1 , \mathbf{t}_2 , and \mathbf{t}_0 layer displacement translations $W\mathbf{t}_1$, $W\mathbf{t}_2$, and $W\mathbf{t}_0$, respectively, which can be determined by careful simulation of an experimental XRD pattern. Based on this model, Plançon *et al.* (1989) successfully simulated a set of experimental XRD patterns of natural kaolinite samples and showed that some samples consist of a physical mixture of two distinct populations of kaolinite crystallites, one of which has a low-defect structure and the other a moderate- to high-defect structure. In addition, the model determined by these authors was dominated by C-vacant layers along with a small fraction of B-vacant layers.

Later, Plançon and Zacharie (1990) suggested an alternative approach for the structural characterization of kaolinite based on an expert system which describes the defect structure of kaolinite using a few measurements taken from an experimental powder XRD pattern.

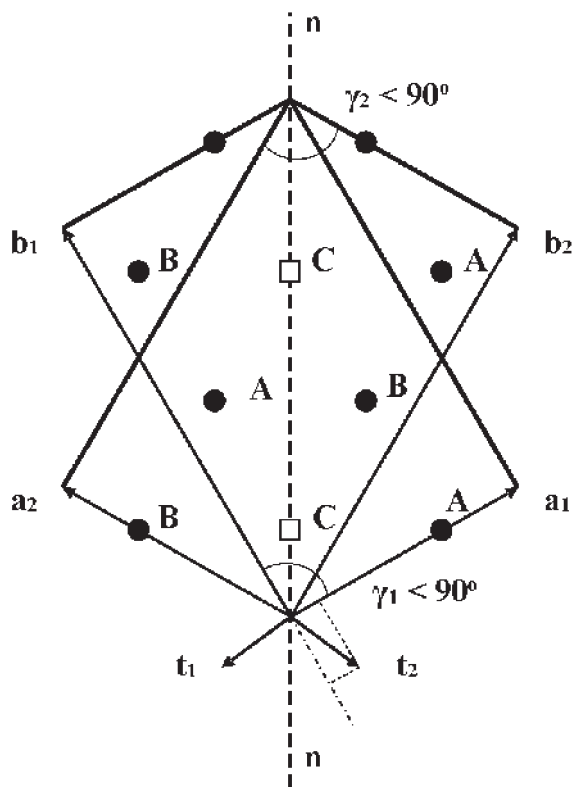


Figure 2. A mutual arrangement of two oblique kaolinite-layer unit cells related by a common pseudo-mirror plane that passes through the centers of vacant C octahedra. The two layer displacements, \mathbf{t}_1 and \mathbf{t}_2 , are related to the corresponding cells by the pseudo-mirror plane.

Application of the expert system to the experimental XRD patterns of nine kaolinite samples whose defect structures were determined by the modeling procedure showed that distinguishing a single phase and quantifying two-phase kaolinite samples is possible.

Interstratification of the right-hand and left-hand structural fragments that consist of the same type of vacant layers in natural kaolinite samples agrees with the results of high-resolution transmission electron microscopy (HRTEM) (Kogure and Inoue, 2005; Kogure *et al.*, 2010; Kogure, 2011). These HRTEM studies showed that the common stacking faults in kaolinite are a disorder of the alternative layer displacement vectors, \mathbf{t}_1 and \mathbf{t}_2 . The two layer displacements are related to each other by a pseudo-mirror plane passing through the vacant octahedral site in the kaolinite unit cell. In addition, application of HRTEM revealed the possible existence of stacking faults related to a mutual layer rotation or to displacement of the octahedral vacancy sites.

Artoli *et al.* (1995) proposed a new single-phase model for describing structural disorder in natural kaolinite samples which was based on the stacking of C-vacant and B-vacant layers and the enantiomorphs. According to these authors, the ordered stacking

sequences of B- and C-enantiomorphic layers represent domains of “monoclinic” kaolinite, like the ordered interstratification of B-layers and C-layers. The four layer types are interstratified and the layer distribution can be described for any total density of stacking faults.

Recently, Ufer *et al.* (2015) applied the Rietveld method for a quantitative phase analysis of kaolinite and pyrophyllite samples. In particular; they showed that the KGa-2 kaolinite sample consists of a single, intermediately disordered phase, whereas, the KGa-1 kaolinite sample is a physical mixture of highly (71.1%) and moderately (28.9%) disordered phases. A similar quantification was carried out for a moderately and a highly disordered pyrophyllite. The potential advantages and limitations of the Rietveld method for mineral quantification of coexisting dioctahedral minerals were also discussed (Ufer *et al.*, 2015).

CALCULATION OF POWDER XRD PATTERNS FOR DEFECT STRUCTURAL MODELS OF KAOLINITE

Experimental

Random powder X-ray diffraction patterns from KGa-1 and KGa-2 bulk samples were obtained with a Bruker D8 diffractometer (Bruker AXS, Karlsruhe, Germany (40 kV, 40 mA, Bragg-Bretano mode with a 250 mm goniometer radius) for KGa-1 and KGa-2 and with a Rigaku Smartlab® diffractometer (Rigaku Corporation, Tokyo, Japan) for KGa-1b using $\text{CuK}\alpha$ radiation and a graphite crystal monochromator from $10\text{--}65^\circ 2\theta$ with a $0.05^\circ 2\theta$ step increment and a count time of 100 s/step.

Methodology

One effective approach to obtain quantitative information about the actual defect structure of a fine-grained, dispersed mineral is by modeling the diffraction effects to provide the best agreement between positions, intensities, and profiles of the calculated and observed experimental *hkl* reflections in XRD patterns (Drits and Tchoubar, 1990). To simulate the diffraction effects for kaolinite structural models, the atomic coordinates of the defect-free kaolinite structure refined by Bish and von Dreele (1989) were used with a computer program of Sakharov and Naumov, which is based on algorithms described by Drits and Tchoubar (1990). The main advantages of the program, which was applied for the first time to the study of the kaolinite structure, is that the intensity distribution is calculated for the whole XRD pattern including both $00l$ and *hkl* reflections using the same set of structural parameters. Model structures may contain an arbitrary number of different types of defects and the distribution of the defects may obey an arbitrary value of a short-range order factor. The program can minimize discrepancies between the experimental and calculated XRD patterns.

X-ray diffraction patterns were calculated for kaolinite structural models consisting of a statistically homogeneous population of crystallites, each of which has a given content of the randomly interstratified layer displacement vectors and a set of structural parameters described below. In the calculated models, the crystallites or coherent scattering domains (CSDs) in the *ab* plane have a disk-like shape and different radii, each of which occurs with equal probability. The CSD thicknesses were distributed log-normally along the c^* axis and the parameters of this distribution were determined using the mean thickness of CSDs, equal to the mean number of kaolinite layers, N , multiplied by the layer thickness, and the regression given by Drits *et al.* (1997b). Corrections for instrumental factors, such as horizontal and vertical beam divergence, length, width, and sample thickness were carried out according to the recommendations of Drits *et al.* (1993) and Reynolds (1986). A model of the defect kaolinite structure consisting of the same layer type was defined by the probability of t_1 , t_2 , and t_0 translations Wt_1 , Wt_2 , and Wt_0 . If the layer displacement vectors are not fixed in length, the variation of the vectors around the average values is described by Gaussian distributions, where δ_x , δ_y , and δ_z are, respectively, the characteristic widths (in fractions of the unit cells) of the distribution along the *a*, *b*, and *c* directions.

Powder XRD patterns from kaolinite samples may differ because of different mutual orientations of kaolinite particles. The degree of particle orientation was estimated by an angular parameter α equal to the full width at half of a particle distribution (Drits and Tchoubar, 1990). For each particular model, two XRD patterns were calculated using $\text{CuK}\alpha_1$ and $\text{CuK}\alpha_2$ wavelengths and were summed in the ratio of 2:1, respectively. When the structural parameters for each model are fixed, the program automatically seeks the best agreement (minimum R_p factor) between the experimental and calculated XRD patterns by varying the content of each population in a sample. The same program was used to provide a maximum possible agreement between the XRD pattern from a sample consisting of a mixture of two phases named as N and M, respectively, and a XRD pattern simulated as a sum of XRD patterns corresponding to the N and M phases. In this case, the content of N and M phases in the studied sample should be equal to contributions of the XRD patterns of the N and M phases in the XRD pattern of the sample.

Layer displacements used for XRD pattern modeling

Application of the trial-and-error approach showed that the best possible agreements between the experimental and calculated d_{hkl} values can be achieved when the unit cell parameters are: $a = 5.1575 \text{ \AA}$, $b = 8.9474 \text{ \AA}$, $c = 7.1545 \text{ \AA}$, $\alpha = \beta = 90^\circ$, and $\gamma = 89.841^\circ$ and the layer displacement vector t_1 is equal to:

$$\mathbf{t}_1 = -0.36815\bar{a} - 0.0225\bar{b} + 7.1545\bar{n}$$

where \bar{n} is a unit-cell vector along the c^* axis. This set of the unit-cell parameters differs slightly from that determined by Bish and von Dreele (1989) for the Keokuk sample:

$$a = 5.1554 \text{ \AA}, b = 8.9448 \text{ \AA}, \gamma = 89.922^\circ, \\ \mathbf{t}_1 = -0.36828\bar{a} - 0.0239\bar{b} + 7.1539\bar{n}$$

The second layer displacement vector, \mathbf{t}_2 , is obtained from \mathbf{t}_1 by symmetry to the pseudo-mirror plane of C-vacant layers that passes through the centers of the vacant octahedral sites. As was mentioned, the components of \mathbf{t}_2 on the a_1 and b_1 axes are close to $a_1/6$ and $-b_1/6$, respectively (Figure 2). Careful calculations showed that these values are $0.1500a_1$ and $-0.1953b_1$. Taking into account that $a_1 = a$, and $b_1 = b$ and the unit cell is base-centered, the values for the \mathbf{t}_2 translation are:

$$\mathbf{t}_2 = -0.3500\bar{a} + 0.3047\bar{b} + 7.1545\bar{n}$$

One can show that the unit-cell parameters determined for the studied samples do not significantly change interatomic distances compared to those obtained using the unit-cell parameters of Bish and von Dreele (1989).

The third layer displacement vector, \mathbf{t}_0 , is placed along the long diagonal of the unit-cell axis located on the pseudo-mirror plane:

$$\mathbf{t}_0 = -0.3154\bar{a} - 0.3154\bar{b} + 7.1545\bar{n}$$

The modulus of \mathbf{t}_1 , \mathbf{t}_2 , and \mathbf{t}_0 are equal to each other. Thicknesses of crystallites or coherent scattering

domains, CSDs were determined by modeling the profiles of the 001 and 002 reflections. The mean and maximum thicknesses of CSDs having a log-normal distribution are used as variable parameters. Interstratification of \mathbf{t}_1 and \mathbf{t}_2 layer displacement vectors have a minimal influence on the positions and intensity distribution in the diffraction region containing the triplet of strong maxima having $\bar{1}30$, $\bar{1}\bar{3}1$, and 200 indices. The profiles of these reflections in the experimental XRD patterns were, therefore, initially modeled for the defect-free structure consisting of C-vacant layers. This approach allowed estimation of the radius of the disks in the CSDs and the distribution of discs in the ab plane.

SIMULATION OF THE EXPERIMENTAL XRD PATTERNS

KGa-1 sample

Determination of the preliminary estimated parameters allows XRD pattern simulation corresponding to a defect-free model and a model in which layer displacement vectors \mathbf{t}_1 and \mathbf{t}_2 occur with equal probability and in the same proportion. The comparison of these patterns with that of the KGa-1 sample (Figure 3) indicates at a qualitative level that the sample consists of two phases that differ from each other by the content of stacking faults, where the peak positions of the 021 and 111 reflections in the experimental XRD pattern and that of the defect-free model almost coincide. This indicates that the highly ordered population of crystallites in the sample named the HOK phase,

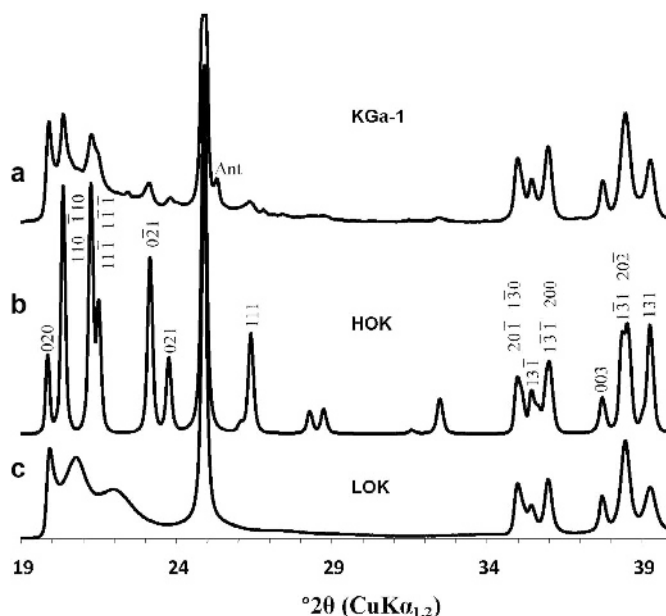


Figure 3. Experimental XRD pattern (a) compared with XRD pattern calculated for the defect-free kaolinite structure (b), and with XRD pattern calculated for the structure with equal amounts of randomly interstratified enantiomorphous kaolinite fragments (c).

contains only a few stacking faults (Figure 3). The positions of the 11/ and 02/ reflections are very sensitive to the relative contents of the layer displacement vector t_2 . Calculations showed that an increase of t_2 layer displacements above 5% noticeably shifted the positions of the 02/ and 11/ reflections in XRD patterns of the HOK phase relative to those observed in the experimental XRD pattern. The significant background separating individual 02/ and 11/ reflections from the base line of the experimental XRD pattern probably are due to the contribution of a low ordered population of crystallites named the LOK phase.

The defect structure of each of the coexisting phases in the sample was determined by a trial and error method by variation in defect types, defect occurrence, and the content of the phases. The maximum possible agreement between the experimental and modeled powder XRD patterns of the KGa-1 sample was achieved using a set of parameters that described the defect structural models of the HOK and LOK phases (Table 1). The HOK phase has an almost defect-free structure in which 97% of the layer pairs are related by the layer displacement vector t_1 and only 3% of the layer pairs form enantiomorphic fragments. In contrast, the LOK phase has three t_1 , t_2 , and t_0 layer displacements that occur in ratios 0.55:0.35:0.05 and are interstratified at random.

Two distinct features exist between the HOK and LOK the phases. First, the LOK phase contains 5%

stacking faults due to layer displacement by t_0 . This kind of defect slightly improves the quality of the agreement between the experimental and calculated XRD patterns. Model calculations show, even visually, that the agreement between the experimental and calculated XRD patterns is worse when the LOK phase has no stacking faults from t_0 layer displacements (not shown). The second difference between the probability parameters describing the coexisting phases is the presence of 5% of arbitrary stacking faults, W_a , in the LOK phase which displace adjacent layers with respect to each other in arbitrary directions and lengths (details in Drits and Tchoubar, 1990).

The XRD patterns corresponding to the HOK and LOK crystallite populations (Figure 4) taken in proportion 24.3% and 75.7%, respectively, provides the optimal agreement between the experimental and calculated XRD patterns at $R_p = 8.12\%$. This result shows that the sample consists of a physical mixture of 24.3% HOK and 75.7% LOK phases. The proportion of the LOK to HOK phase determined in this work is close to that of similar phases in the KGa-1 sample (29:71) quantified by the Rietveld method (Ufer *et al.*, 2015). Three fragments of the experimental XRD pattern and patterns of the HOK and LOK phases presented in a larger scale shows that the HOK crystallites contribute significantly to the positions, widths, and intensities of the 02/ and 11/ reflections, whereas the LOK crystallites

Table 1. Structural parameters of HOK and LOK fractions in KGa-1, KGa-1b, and KGa-2.

Unit cell parameters of kaolinite phase	a (Å)	b (Å)	c^* (Å)	γ (deg)	t_1	t_2	t_0
HOK	5.1575	8.9474	7.1545	90	$-0.3681a - 0.0225b$	$-0.3499a + 0.3047b$	$-0.3154a - 0.3154b$
LOK	5.1575	8.9474	7.1545	90	$-0.3681a - 0.0225b$	$-0.3499a + 0.3047b$	$-0.3154a - 0.3154b$

Sample Phase	— KGa-1 —		— KGa-1b —		— KGa-2 ₁ —	
	HOK	LOK	HOK	LOK	HOK	LOK
$W(t_1)$	0.97	0.55	0.97	0.55	0.97	0.55
$W(t_2)$	0.03	0.35	0.03	0.38	0.03	0.35
$W(t_0)$	—	0.05	—	0.05	—	0.05
W_a	—	0.05	—	0.02	—	0.05
$\delta(t)$	0.05c	0.02a, 0.02b, 0.05c	0.05c	0.02a, 0.02b, 0.05c	0.03a, 0.02b, 0.05c	0.02a, 0.02b, 0.05c
N	45	45	45	45	25	25
D (Å)	200–800	200–800	200–800	200–800	200–800	200–500
α (deg)	80	80	180	180	150	150
C (%)	24.3	75.7	28.4	71.6	4.0	96.0

W_a is the amount of arbitrary stacking faults which displace adjacent layers with respect to each other in arbitrary directions and lengths (see details in Drits and Tchoubar, 1990), N is the mean number of layers in coherent scattering domain (CSD), D (Å) is CSD diameter, α is a factor estimating the degree of particle orientation.

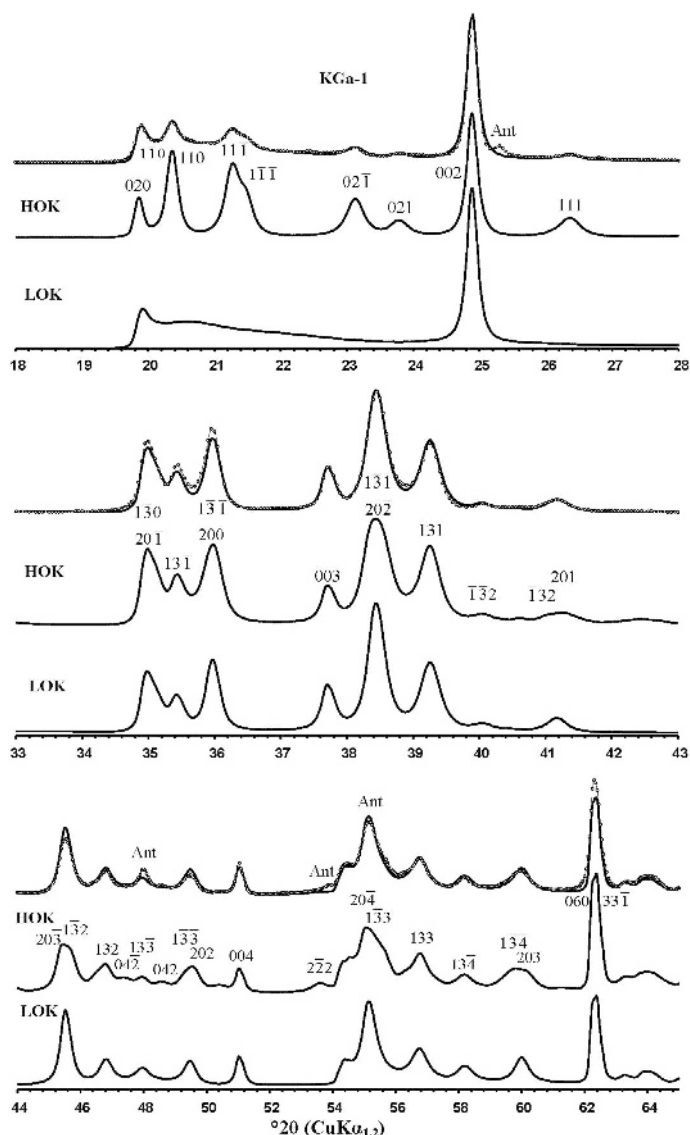


Figure 4. Most diagnostic parts of experimental KGa-1 XRD pattern (dotted line) showing agreement with patterns obtained by optimal summation of the calculated XRD patterns (solid line) of the HOK and LOK components.

contribute to the background in a non-uniform manner along with the profile of the 020 reflection (Figure 4).

KGa-1b sample

Modeling of the experimental XRD pattern of the KGa-1b sample showed that the phase composition of the sample is similar to that of KGa-1. Both samples consist of HOK and LOK phases, and both have the same unit cell parameters and identical sets of probability parameters (Table 1). The HOK phases in both samples have the same ratios of the t_1 and t_2 layer displacements, whereas, in the LOK phase, the relative contents of layer displacements t_2 , t_0 , and arbitrary stacking faults are equal to 0.38, 0.05, and 0.02 instead of 0.35, 0.05, and 0.05 determined for the LOK of the KGa-1 sample

(Table 1). The XRD patterns corresponding to the HOK and LOK crystallite populations (Figure 5) taken in proportions of 28.4% and 71.6%, respectively, provided the optimal agreement between the experimental and calculated patterns at $R_p = 9.74\%$. The main difference between the KGa-1b and KGa-1 samples is, thus, that KGa-1b contains a higher content of the HOK phase (28.4%) compared with KGa-1 (24.3%).

KGa-2 sample

The best possible agreement between the experimental and modeled powder XRD patterns (Figure 6) was based on a set of probability parameters that described the defect structural model of the HOK and LOK phases (Table 1) from modelling the whole XRD pattern. The

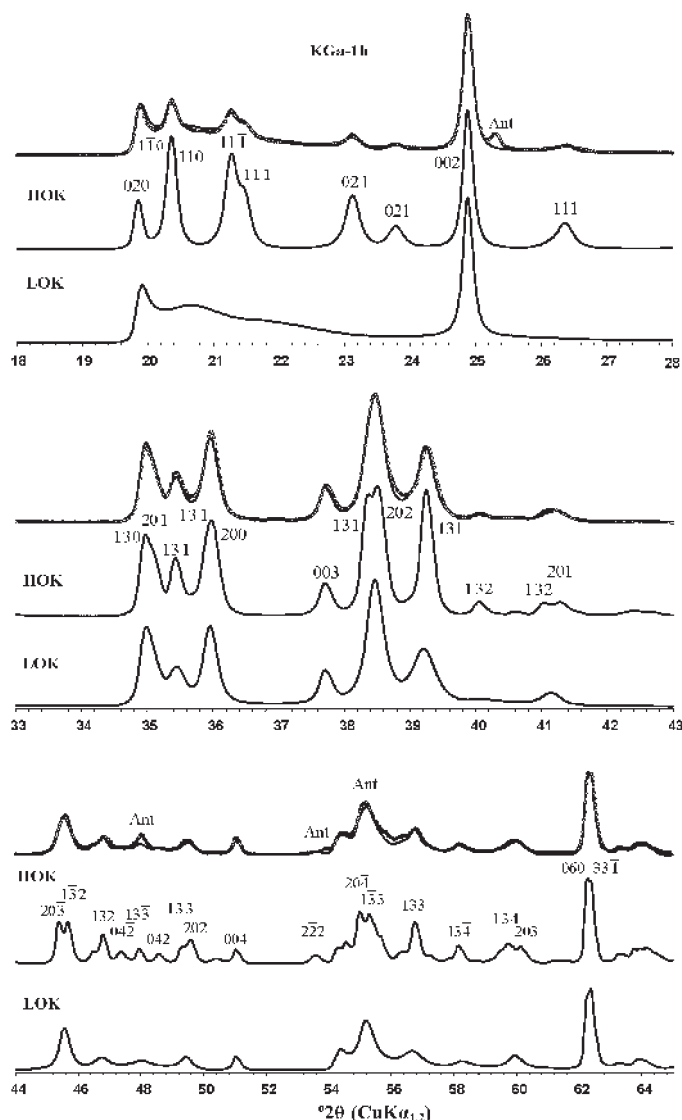


Figure 5. Most diagnostic parts of an experimental KGa-1b XRD pattern (dotted line) showing agreement with patterns obtained by optimal summation of the calculated XRD patterns (solid line) of the HOK and LOK components.

results of the modeling show that the KGa-2 sample also consists of two populations of crystallites with the same unit cell parameters as those found for the HOK and LOK phases of KGa-1 and KGa-1b. In one of the populations, crystallites have an almost defect-free structure identical to that determined for the KGa-1 and KGa-1b samples. In this phase, 97% of the layer pairs are related by the layer displacement vector t_1 and 3% of the layer pairs form enantiomorphic fragments. As in the LOK phases of KGa-1 and KGa-1b, this phase along with the high amount of t_1 (55%) and t_2 (35%) interlayer displacements contains 5% of the layer displacement t_0 and 5% of arbitrary stacking faults (Table 1). The best agreement between the model and experimental XRD patterns was achieved with the HOK

and LOK phases in the proportion 0.96:0.04 with $R_p = 10.8\%$ (Figure 6).

Results obtained by expert system of Plançon and Zacharie (1990)

Application of the expert system to the XRD patterns of the KGa-1, KGa-1b, and KGa-2 samples showed that a monophasic defect structure of KGa-2 can be distinguished from a two-phase structure of KGa-1 and KGa-1b. The expert system and modeling results from this study showed that the kaolinite samples consist of only C-vacant layers. In addition, for the KGa-1 and KGa-1b samples, the proportion of the HOK and LOK phases determined by XRD modeling (24.4:75.6; 28.4:71.6) differed significantly from the ratios found by the expert system (34.3:65.7; 39.9:60.1).

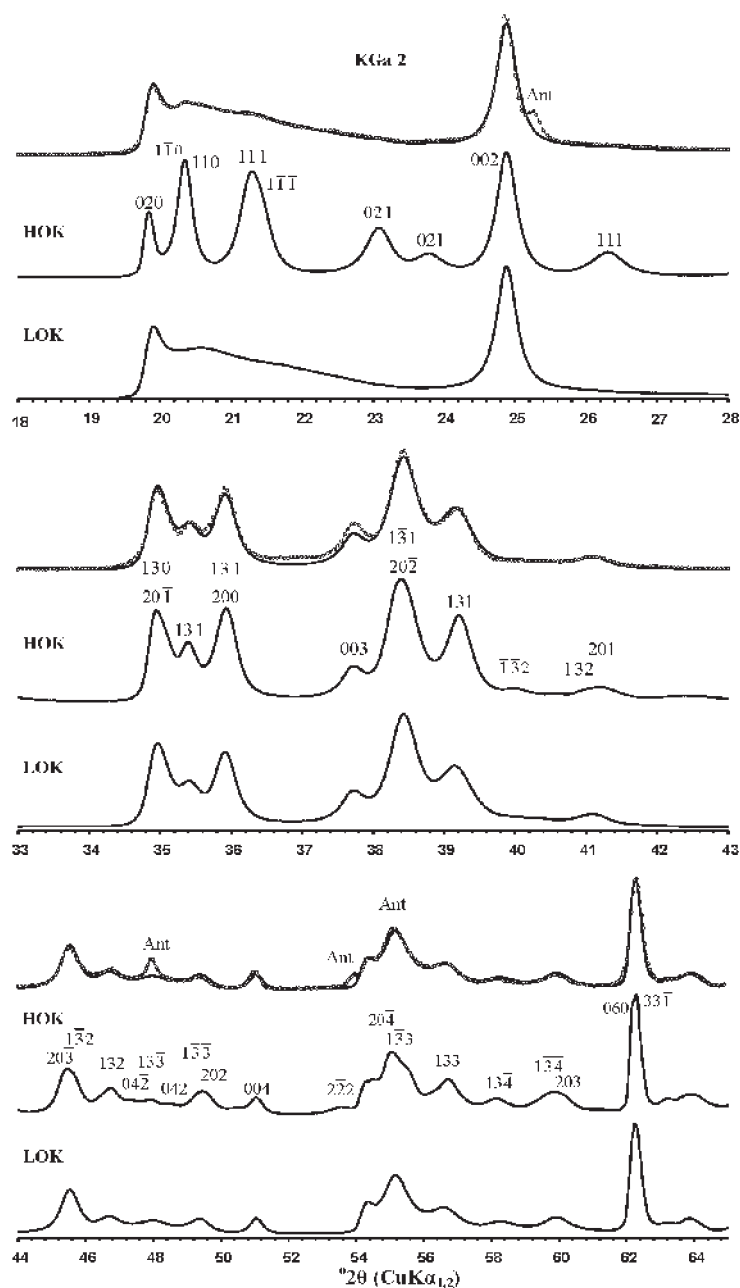


Figure 6. Most diagnostic parts of an experimental KGa-2 XRD pattern (dotted line) showing agreement with patterns obtained by optimal summation of the calculated XRD patterns (solid line) of the HOK and LOK components.

The expert system cannot provide information about structural disorder of the LOK phases in these samples and will not separately distinguish t_2 and t_0 layer displacements. The system will, therefore, give only a global abundance of translation defects, p , equal to the ratio known as the Hinckley Index, HI (Hinckley, 1963; Plançon and Zacharie, 1990). The HI value determined for KGa-2 was 0.22, whereas, the sum of $Wt_2 + Wt_0$ determined by the XRD modelling of this sample was 0.40 (Table 1).

DISCUSSION

The main aim of this work was to determine types of defects and patterns of distribution in the kaolinite structures of the KGa-1, KGa-1b, and KGa-2 samples. The good agreement between the experimental and calculated XRD patterns obtained for the whole $10-65^\circ 2\theta$ diffraction range with low profile fitting factors ($R_p = 8-11\%$) is strong evidence for the validity of the defined defect models (Figure 7). These samples consist of only C-vacant layers and the main source of

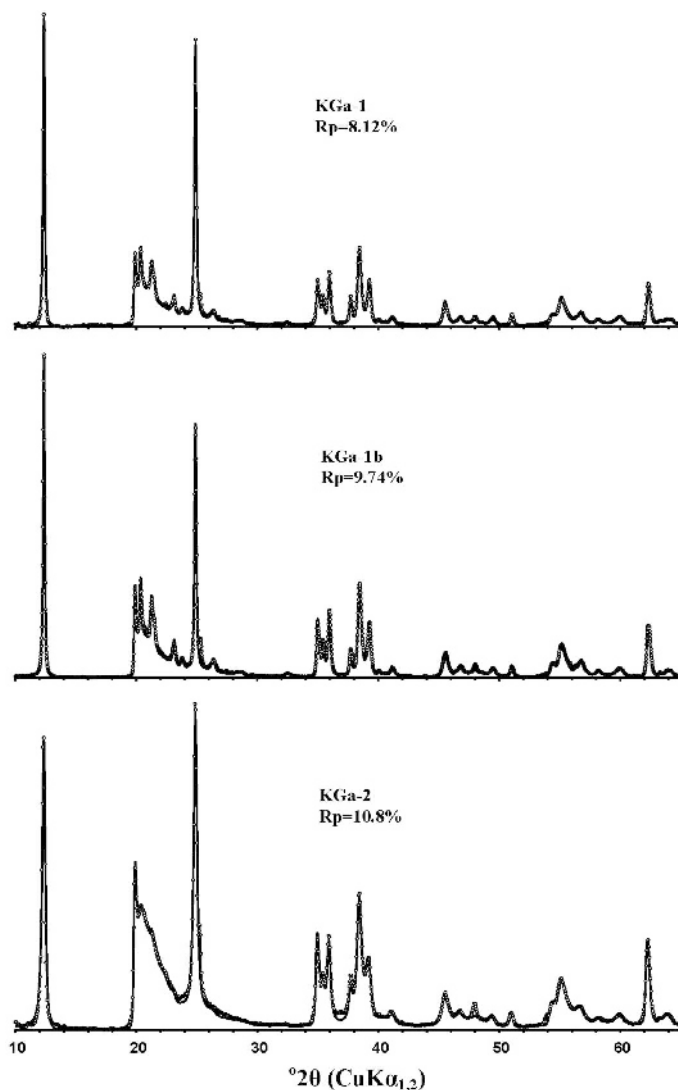


Figure 7. Comparison of the experimental and simulated XRD patterns of KGa-1, KGa-1b, and KGa-2.

defects in the structures is random interstratification of enantiomeric kaolinite fragments related to each other by a pseudo-mirror plane and due mostly to the layer displacement vectors t_1 and t_2 .

The samples contain two populations of crystallites with different contents of stacking faults and were distinguished from each other by the relative amounts of the HOK and LOK phases. Surprisingly, in the so-called well ordered or well crystallized KGa-1 and KGa-1b kaolinites, the relative amounts of the HOK phases were equal only to 24.3 and 28.4%, which are, respectively, 3 and 2.5 times lower than the coexisting LOK phases. Comparison of the XRD patterns of the HOK and LOK phases that compose KGa-1, KGa-1b, and KGa-2 showed that the positions and general shape of the intensity distribution in XRD patterns of the LOK phases were almost identical, which means that KGa-1 and KGa-1b

differ from KGa-2 mostly by the higher content of the HOK phase. The coexistence of HOK and LOK phases in natural kaolinite samples discovered by Plançon *et al.* (1989) is, thus, in full agreement with the results of the present study.

Models for the stacking faults in kaolinite

Defects in crystallites characterized by one layer type.

The symmetrical arrangement of atoms with respect to the pseudo-mirror plane (Figure 1a) suggests a most favorable model for stacking faults. Indeed, assume that the displacement of a layer with respect to the preceding one is described by t_1 . If a regular crystal is formed, all successive layers are shifted by the same vector t_1 . A layer that is related to the previous one by the pseudo-mirror plane, n , creates a stacking fault and the plane acts as a glide plane for these two layers, while both the

periodicity and the cation distribution would remain unchanged. The formation of such a layer would, therefore, lead to only minor changes in the potential energy of layer interaction, as well as in the hydrogen bond energy. The t_2 vector, which is related to t_1 by the same n plane, would be the new translation for a crystal fragment between the “defect” layer and the next stacking fault. The second defect-free fragment thus formed should be enantiomorphic to the first one. Because in the kaolinite layer n and m planes almost coincide (Figure 2), a pair of layers related by the glide plane should coincide with each other. The model of interstratification of enantiomorphic layers, thus, becomes that of alternating identical layers stacked with symmetrical translations t_1 and t_2 . The results of the present work as well as those obtained by modeling of the XRD patterns (Plançon *et al.*, 1989) and HRTEM (Kogure and Inoue, 2005; Kogure *et al.*, 2010, 2011) show that, among the different types of stacking faults, those corresponding to t_1 and t_2 layer interstratification displacements are most abundant. Bookin *et al.* (1989) considered two extreme modes of occurrence of t_1 and t_2 vectors in the defective kaolinite structure consisting of C-vacant layers. The first mode includes interstratification of the enantiomorphic packets each of which has an ordered layer stacking, whereas, in the second mode isolated t_2 stacking faults are inserted occasionally in the ordered matrix with t_1 .

The first mode was first visualized by application of HRTEM to a diagenetic kaolinite sample with images that demonstrated two types of ordered packets having t_1 and t_2 layer translations (Kogure and Inoue, 2005). A similar pattern in the distribution of stacking faults exists in the defect structures of KGa-2 and the LOK phases of KGa-1 and KGa-1b (present work). Indeed, in these structures stacking faults result from the disorder in alternating t_1 , t_2 , and t_0 layer displacements among which t_1 and t_2 occur in commensurable proportions (Figure 8a). This type of stacking fault distribution in the defect kaolinite structure has been called a “growth fault” (Bookin *et al.*, 1989) and “enantiometric twins” (Kogure and Inoue, 2005).

The second type of stacking fault distribution in the defect kaolinite structure was observed by HRTEM in a kaolinite specimen of sedimentary origin (Kogure *et al.*, 2010). In this case, thick periodic packets with t_1 (or t_2) were seldom separated from each other by isolated stacking faults with t_2 (or t_1). The HOK phase calculated for KGa-1 and KGa-1b has this mode of stacking faults. Indeed, interstratification of 3% layer pairs with t_2 and 97% layer pairs with t_1 should provide a defect structural model of the HOK structure in which the layer sequence related by t_1 was seldom interrupted by a single layer with t_2 (Figure 8b).

Stacking disorder by layer rotation and/or displacement of vacant octahedral sites. Application of HRTEM to the

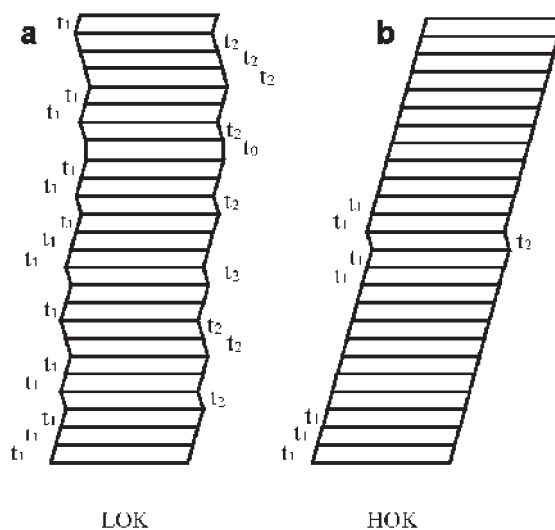


Figure 8. Schematic illustration of two types of structural disorder in KGa-1, KGa-1b, and KGa-2. One type of disorder is from random interstratification of thin enantiomorphic blocks (a) and is similar to KGa-2 and LOK phase structures in which layer displacements t_1 and t_2 occur in compatible proportions (Table 1). The second type of disorder corresponds to the HOK phase because structural periodicity is seldom disturbed by isolated stacking faults (b).

kaolinite sample of sedimentary origin previously studied by Johnston *et al.* (2008) showed that along with interstratification of left- and right-hand structural fragments, the studied kaolinite sample contains another type of stacking disorder related to either mutual layer rotation or to displacement of the vacant octahedral site in the kaolinite layer (Kogure *et al.*, 2010; Kogure, 2011). The geometrical analysis of these stacking faults carried out by Bookin *et al.* (1989) showed that the sequence of kaolinite layers containing stacking faults formed due to layer rotation and the octahedral vacancy displacements cannot be described in terms of the same lattice without mutual layer distortion. This distortion probably increases the energy of the formation of these stacking faults, thus decreasing the occurrence probability.

Potential possibility of additional defects in kaolinite structures

Occurrence of B-vacant layers. The possible coexistence of C- and B-vacant layers in a kaolinite structure should be accompanied by some crystal-chemical restrictions that create the specific features in an XRD pattern. The regular interstratification of C- and B-vacant layers forms the dickite structure in which both layer types have the same displacements along the a axis as that in an idealized ‘monoclinic’ kaolinite structure (Bailey, 1963, 1988). A structural comparison of C-vacant and B-vacant kaolinite layers (Figure 1a,b) shows that the layers are different because the pseudo-mirror planes are

rotated with respect to each other by $\sim 120^\circ$, and the a and b parameters of the unit cells form the acute angle ($\gamma < 90^\circ$) located at the opposite ends of the b parameter common for both cells. In contrast, the a parameter of the cells are slightly inclined toward each other to form the same acute angle γ . The octahedral vacancy in regular kaolinite is, therefore, always near the acute angle γ .

In the dickite structure, because of the ordered interstratification of C- and B-vacant layers, the lateral sizes of layers adjust to each other owing to the transformation of the interstratified oblique unit cells with $\gamma > 90^\circ$ and $\gamma < 90^\circ$ to the common orthogonal layer cell with $\gamma = 90^\circ$. In contrast, irregular interstratification of the layer types having comparable contents should be accompanied by strain from a mutual deformation of the layer lattice periodicities in the BC and CB layer pairs. Due to the minor variation of γ from 90° , however, a small amount of B-vacant layer imperfections may occur in the C-layer matrix. These C layers may also introduce some strain, but the influence on diffraction effects should be negligible.

In order to visualize diffraction effects in coexisting C and B layers, the layer displacements related to different types of the layer pairs must be defined. As was mentioned, the acute angle $\gamma < 90^\circ$ in C-vacant and B-vacant layers should be formed at opposite ends of the b parameter of the unit cells providing incommensurable stacking of CB and BC layer pairs. For a given choice of z direction from tetrahedral to octahedral sheets, the interlayer shift is defined unambiguously by the vacant site in the preceding layer. Indeed, the positions of the basal oxygen atoms of the growing upper layer are predetermined by the positions of OH groups forming the upper basal surface of the lower preceding layer and controlling formation of the most stable interlayer hydrogen bonds. The positions of surface OH groups of the preceding layer are, therefore, predetermined positions of the tetrahedral sheet of the upper layer independent of the positions of vacant octahedral sites in this layer (Figure 1a,b). Thus, the transition from C to B or C layer in the sequence CB or CC is determined by the layer displacement vector \mathbf{t}_1 . In contrast, to carry out the transition from B to C or B layer in the sequence BC or BB layer pairs, the layer displacement vector should have the opposite sign along the y -component, which is:

$$(\mathbf{t}_1)_{BC} = (\mathbf{t}_1)_{BB} = -0.36815\bar{a} + 0.0225\bar{b} + 7.1545\bar{n}$$

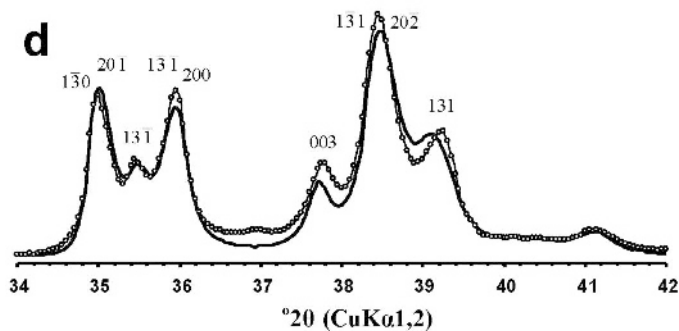
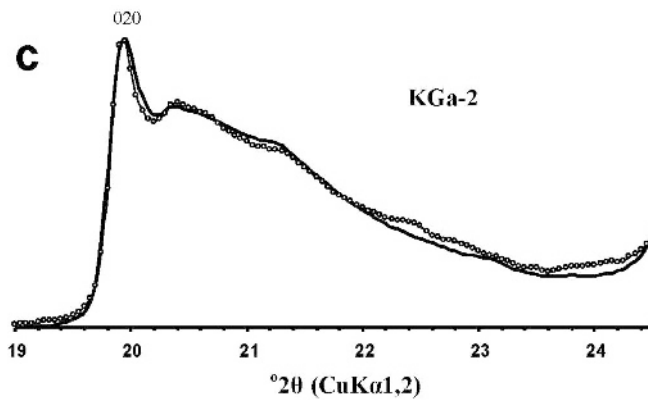
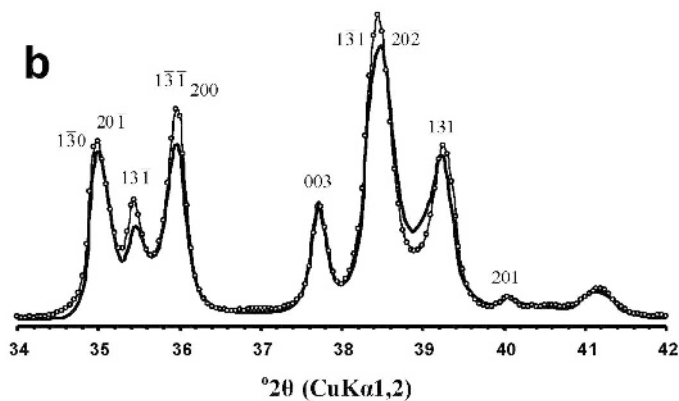
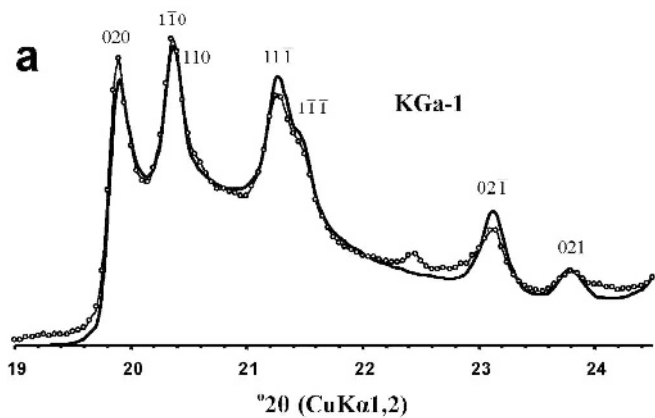
Comparison of the experimental XRD patterns of the KGa-1 and KGa-2 samples with those of their two-phase

models in which the LOK phases contain 10% B-vacant layers shows that the main diffraction effect of B layers is the shift of the 131 reflection to the nearest $1\bar{3}1$ reflection toward a lower 2θ angle along with decreasing resolution (Figure 9b,d). This effect is most clearly seen in the comparison of the experimental and calculated patterns of KGa-2, and is not obvious in the XRD pattern of KGa-1. In order to reveal a small amount of B-layers (<5%) in kaolinite samples with XRD patterns similar to KGa-1, experimental XRD patterns must have high statistics, careful calibration, and accurate instrumental correction factors. Even in this case, the reliable determination of a small amount of B-vacant layers may be ambiguous because similar diffraction effects may be created from layer displacements by \mathbf{t}_0 . Plançon *et al.* (1989), however, determined a coexisting small amount of B layers among C layers and Kogure *et al.* (2010) admitted the existence of local dickite-like fragments formed by layer subsequences C-B-C in a particular kaolinite sample of sedimentary origin which had an IR spectrum that contained dickite bands (Johnston *et al.*, 2008).

Layer displacements by $\mathbf{t}_1 - b/3$. Among the possible stacking faults in kaolinite structures are effects that originate due to the layer displacement vector $\mathbf{t}_1 - b/3$ (Table 1). Brindley (1980) suggested these types of stacking faults by assuming that such displacement does not disturb the mutual arrangement of basal oxygen and hydroxyls of adjacent layers. In the case of the interlayer shift by $-b/3$, however, the octahedral cations of the lower layer should be located exactly under the tetrahedral Si atoms of the upper layers. As noted by Newnham (1961) and supported by the electrostatic energy calculations of Giese (1982), this arrangement is energetically unfavorable. The origin of the $\mathbf{t}_1 - b/3$ layer displacement, however, is possibly related to post-crystallization mechanical effects (Bookin *et al.*, 1989).

Comparison of the experimental XRD patterns obtained in this study with those calculated for structural models in which the LOK phases contain 10% of the $\mathbf{t}_1 - b/3$ layer displacements shows that at this stage of modeling, this amount of these stacking faults contributes selectively to the different parts of the XRD patterns. Indeed, the mutual displacement of the adjusted layer by $b/3$ along the b axes does not influence the positions and intensities of 201 and 131 reflections (not shown). In contrast, this type of stacking fault disturbs the positions and intensities of 021 and 111 reflections (Figure 10). In general, a model containing a small amount of B-vacant layers, $\mathbf{t}_1 - b/3$ stacking faults, or

Figure 9 (*facing page*). Parts of experimental KGa-1 XRD patterns showing agreement with reflections in the 021, 111 (a) and 201, 131 (b) regions for patterns obtained by the optimal summation of calculated HOK and LOK components, where LOK contains 10% B-vacant layers. Parts of the experimental KGa-2 XRD patterns showing agreement with reflections in the 021, 111 (c) and 201, 131 (d) regions for patterns obtained by the optimal summation of calculated HOK and LOK components, where LOK contains 10% B-vacant layers.



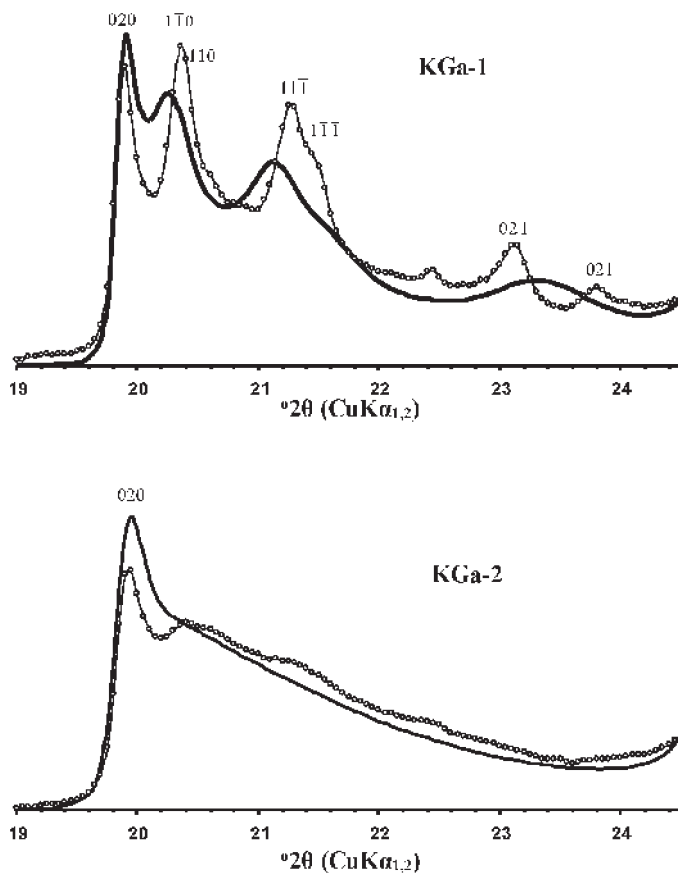


Figure 10. Parts of experimental KGa-1 and KGa-2 XRD patterns containing reflections in the 021, 111 region compared with patterns calculated for a model in which the HOK and LOK components of both samples contain 10% layer displacements of $t_1 - b/3$ instead of the t_1 and t_2 displacements.

other unusual defects that can provide a reasonably good fit with the experimental XRD pattern cannot be excluded. Variable parameters which have clear crystal-chemical meaning, however, are preferred to describe realistic structure models.

Stacking disorder by layer rotation and/or displacement of vacant octahedral sites. Application of HRTEM to the kaolinite sample of sedimentary origin previously studied by Johnston *et al.* (2008) showed that along with interstratification of left- and right-hand structural fragments, the studied kaolinite sample contains another type of stacking disorder related to either mutual layer rotation or to displacement of the octahedral vacancy site in the kaolinite layer (Kogure *et al.*, 2010, Kogure, 2011). The geometrical analysis of these stacking faults carried out by Bookin *et al.* (1989) showed that the sequence of kaolinite layers containing stacking faults formed due to layer rotation or the octahedral vacancy displacements cannot be described in terms of the same lattice without mutual layer distortion. This distortion probably increases the energy of the formation of these stacking faults and decreases the occurrence probability.

Relationships between structural disorder recorded in a particle population and CSDs in the population.

According to definition, a CSD is formed by a set of strictly parallel layers providing coherent scattering of X-rays, neutrons, and electrons. Individual particles are usually an aggregate of CSDs which may have defect-free as well as similar or different defect structures. An individual particle, therefore, is characterized by statistically weighted structural disorder. Particles of a sample usually have a size distribution and with increasing size, the structural disorder of particles may either increase, decrease, or remain the same depending on the structural features of the CSDs that compose each particle-size fraction (Bish and Chipera, 1998).

To illustrate, let us consider the variation of the phase composition and structural features as a function of particle size for KGa-1 and KGa-2 as determined by Drits and Derkowski (2015). The KGa-2 sample has the same phase composition and structural disorder independent of the particle-size fraction. Particles of different sizes have the same proportion of the HOK and LOK phases and the same XRD characteristics.

KGa-1 has a bimodal particle-size distribution with a coarser fraction of lower structural order albeit higher CSDs or crystallite thickness, whereas the finest size fraction has higher stacking order and is associated with thinner crystallites. Finally, the finer fractions of KGa-1 particles contain a higher proportion of the HOK phase compared with the bulk sample. Predicting the relationship between the structural and compositional features of each particle size fraction in advance is, therefore, difficult even if the phase composition and structural features of CSDs that compose a sample are known.

Reappraisal of kaolinite defect models suggested in the literature

Artioli et al. (1995) model. This model describes a structural disorder in natural kaolinites based on the stacking of B- and C-vacant layers and the enantiomorphs. Here the structure of each crystallite contains four layer displacement vectors t_1 , t_2 , t'_1 , and t'_2 with occurrence probabilities derived from the probability recursive matrix. According to this model, a defect structure of well and poorly crystallized kaolinite with diffraction features similar to KGa-1 and KGa-2 can be described in terms of a single phase structural model; however, the incommensurability that should occur in BC and CB layer sub-sequences where comparable amounts of C layers and B layers coexist was not considered. The authors applied this approach to the kaolinite sample of hydrothermal origin named KAOSAR, which has an XRD pattern similar to the KGa-1 and KGa-2 standards. Even a visual observation of figure 9 in Artioli *et al.* (1995) demonstrates a strong disagreement between the experimental and modeled XRD patterns. Taking into account the crystal-chemical problems related to incorporating C- and B-vacant layers and the enantiomorphs, along with the poor calculated fit with the experimental data, this model can be rejected.

What is the highest probability of layer displacements t_0 that can exist in kaolinite structure? In this study, special attention was paid to see the influence of t_0 translation on the diffraction effects because of recent publications by White *et al.* (2013a,b). These authors showed that vibration dynamics of the crystal structure of natural kaolinite, KGa-1b, calculated using the density functional theory, agrees well with the experimental inelastic neutron scattering (INS) spectrum except for distinct discrepancies in the low frequency region (200–400 cm^{-1}). Analysis of possible types of stacking faults in the kaolinite structure, carried out by these authors, showed that the agreement between the calculated and observed low frequency regions are improved through the emergence of longer hydrogen bonds (O–H \cdots O) that occur due to the formation of stacking faults from the layer displacement vector t_0 . The residual disagreement between the experimental and calculated spectra in the low frequency region was

attributed to quantum tunneling of the inner surface H atoms associated with the hydrogen bonds (O–H \cdots O). This effect should lead to a softening of the low frequency modes in the experimental spectrum compared to the calculated spectrum.

White *et al.* (2013a) concluded that the layer displacements by t_0 are the dominant stacking fault in KGa-1b kaolinite because this defect resolved the discrepancy between experiment and theory for the H-atom dynamic in kaolinite. Because of the lack of direct evidence of layer displacements by t_0 in the kaolinite structure, a set of calculations was carried out in this study to support or reject a t_0 based model. In particular, XRD patterns for models of KGa-1b and KGa-2 samples in which the LOK phases contain different contents of these stacking faults were calculated. Even a model with 15% of the t_0 layer displacements in the LOK phase of KGa-2 dramatically changed the positions and intensity distribution in both the 02l, 11l and 20l, 13l regions in the simulated XRD patterns compared with those in the experimental patterns (Figure 11a,b). Similar discrepancies were observed between the experimental XRD pattern and that calculated for a model in which the LOK phase of KGa-1 contains 20% of the t_0 stacking faults (Figure 11c,d).

The positions of 02l and 11l reflections in the experimental XRD pattern of KGa-1b almost coincide with those of the HOK phase showing that this phase is almost defect free and has a structure close to that of the periodic 1Tc kaolinite (Figures 3, 4). In contrast, the positions and intensity distribution of *hkl* reflections of the periodic structure formed by t_0 layer displacements differed dramatically from those in the experimental XRD pattern (Figure 12). The obvious reason is that layer sequences whose formation is governed either by t_1 or t_0 layer displacements have principally different structures and, therefore, different positions and intensities of *hkl* reflections. Indeed, the direction of the t_0 layer displacements coincides in projection on the *ab* plane with the long diagonal of the oblique kaolinite unit cell of the C-vacant layers. The degree of disagreement between the compared XRD patterns increases strongly when t_0 layer displacements are introduced into the HOK structure (Figure 12). The layer sequence should, therefore, correspond to a monoclinic unit cell (Bailey, 1988). Indexing of *hkl* reflections in the XRD pattern calculated for a model in which layer displacements are governed only by t_0 (Figure 12) allowed determination of a new set of unit cell parameters: $a = 5.1805 \text{ \AA}$, $b = 8.9435 \text{ \AA}$, $c = 7.4069 \text{ \AA}$, and $\beta = 105.0^\circ$, which provide a good agreement between the observed and calculated d_{hkl} values.

Thus, the results obtained in the present paper show that the conclusion of White *et al.* (2013a,b) concerning the nature of the dominant type of stacking faults does not correspond to the actual structure of KGa-1b kaolinite. Therefore, the theoretical problem related to

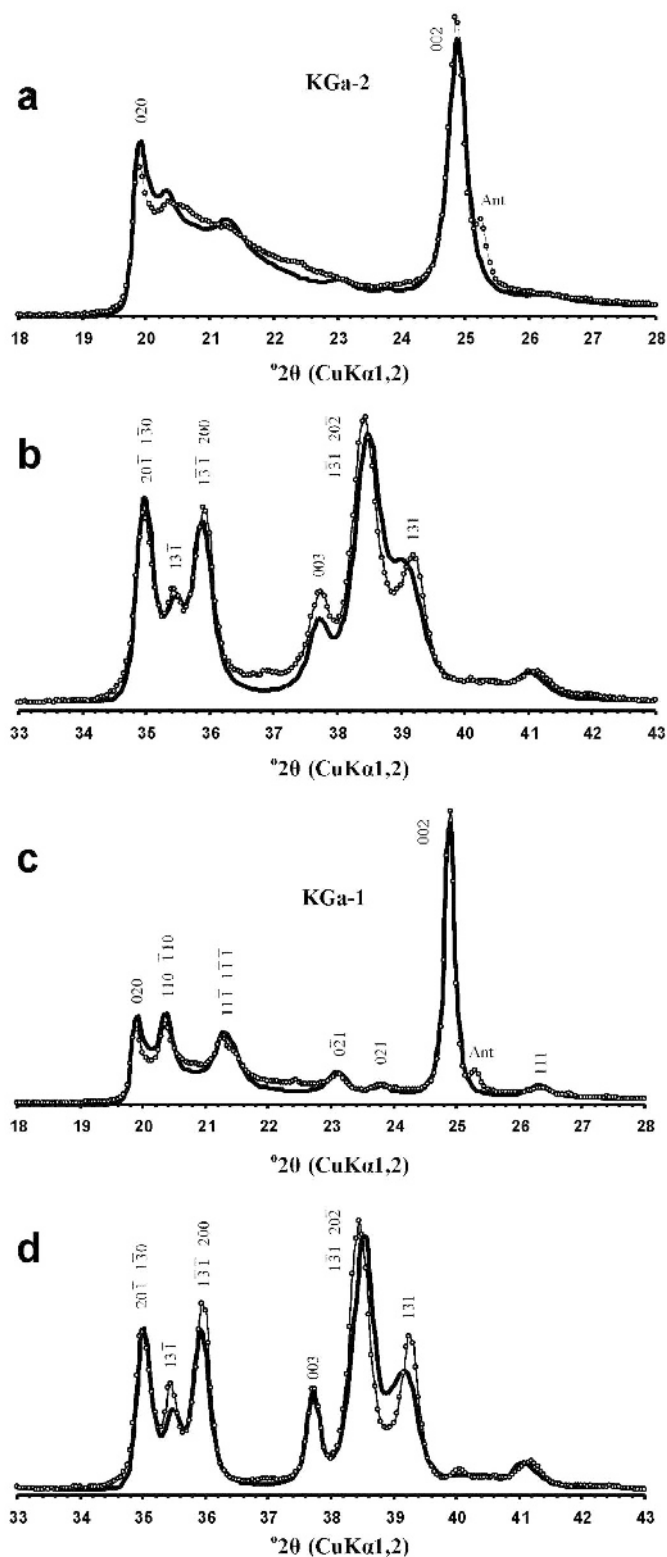


Figure 11. Comparison of the 02 l , 11 l (a) and 20 l , 13 l (b) regions in the experimental KGa-2 XRD patterns (dotted lines) with patterns obtained by optimal summation of calculated XRD patterns of the HOK and LOK phases (solid lines) in which the content of t_0 stacking faults in the LOK phase was increased from 5 to 15%. Comparison of the 02 l , 11 l (c) and 20 l , 13 l (d) regions in the experimental KGa-1 XRD patterns (dotted lines) with patterns obtained by optimal summation of calculated XRD patterns of the HOK and LOK phases (solid lines) in which the content of t_0 stacking faults in the LOK phase was increased from 5 to 20%.

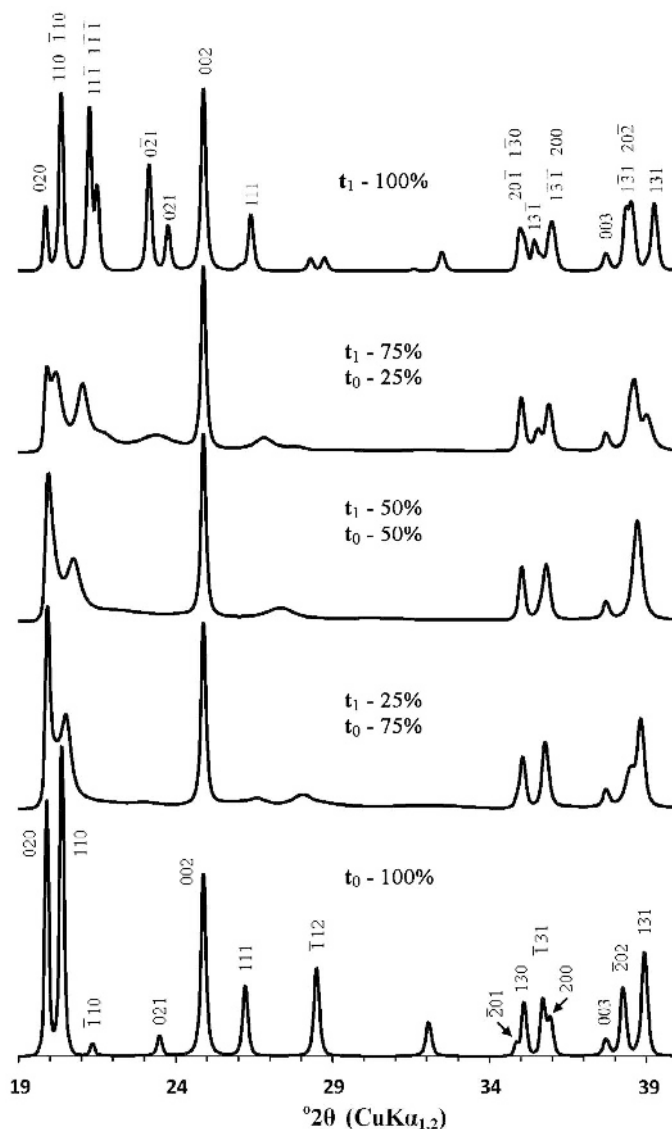


Figure 12. The evolution of positions and intensities of 02l, 11l, 20l, and 13l reflections in the XRD patterns calculated for models in which the contents of the interstratified t_0 and t_1 layers change from 0 to 100%.

interpretation of the low frequency modes in the experimental INS spectra of kaolinite samples requires further investigation.

Why knowledge of the actual kaolinite structures may be useful for interpretation of experimental data? As was mentioned, Paris (2014) discovered that two aluminum sites in the ^{27}Al MAS NMR spectrum of KGa-1b have different spectral parameters. According to the author, although the spectrum decomposition was very accurate, the experimental NMR parameter values must be justified independently, which was the actual problem. The choice of the KGa-1b sample was based on a firm belief that the sample contains a low defect content.

Therefore, to confirm the validity of the spectral parameters obtained for two aluminum sites, the kaolinite structure with 3D periodicity was chosen for first-principle calculations of the NMR parameters. As was shown in the present study, the KGa-1b sample has a 28.4% defect-free phase; where 71.6% of the phase has a comparable amount of right- and left-hand kaolinite structural fragments that are interstratified at random. The application of first-principle calculations of the NMR parameters from the periodic kaolinite structure cannot, therefore, be considered as independent evidence of the validity of the experimental spectroscopic data obtained from the sample with a high content of stacking faults.

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